Light Trapping in Single Coaxial Nanowires for Photovoltaic Applications

W. F. Liu, J. I. Oh, and W. Z. Shen

Abstract—We report a strong enhancement of the light absorp-5 tion in single coaxial nanowires (NWs) of Si core/dielectric shells. 6 We have calculated the light absorption coefficient within the 7 framework of the Lorenz-Mie light scattering theory and found 8 out that it is greatly increased by effective light trapping in Si 9 cores owing to dielectric shells, as compared to that in Si NWs. 10 We show that the strong absorption of light stems mainly from 11 off-resonance enhancement and also from resonance contribution. 12 By optimally tuning the core radius, the shell thickness, and the 13 shell refractive index, we have obtained $\sim 102\%$ increase of the 14 photocurrent.

15 Index Terms—Fano effect, leaky-mode resonances (LMRs), 16 light trapping, off-resonance, single coaxial nanowires.

I. INTRODUCTION

18 IGHT trapping is a powerful means to enhance the light 19 ▲ absorption of solar cells [1]–[5]. In commercial crystalline 20 Si solar cells, light trapping is typically realized by using 21 the pyramidal textured surface to increase the effective path 22 length of light in the cells [1]. Plasmonics, an emerging field 23 for guiding and localizing light at subwavelength scale, has 24 been becoming a new method for light trapping in thin film 25 solar cells [2]. For nanowire (NW) solar cells, particularly Si 26 NWs with radial p-n junctions [6], although they have been 27 thought to reduce both required quality and quantity of Si due 28 to their intrinsic structure that will orthogonalize the directions 29 of light absorption and charge collection [7], [8], effective 30 light trapping in NW solar cells has not been much studied to 31 date. Recently, there have been some effective light trapping 32 techniques reported in Si microwire arrays [3] and in ordered 33 Si NW arrays [4]. However, light trapping in single NWs still 34 remains unexplored.

35 It has been shown [5] that one can engineer the resonant 36 property inside *single* NWs by tuning the radius so that the

Manuscript received September 21, 2010; accepted October 1, 2010. This work was supported in part by the National Major Basic Research Project under Grant 2010CB933702 and in part by the Natural Science Foundation of China under Contracts 10734020 and 11074169. The review of this letter was arranged by Editor P. K.-L. Yu.

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Digital Object Identifier 10.1109/LED.2010.2086428

light absorption can be enhanced at resonance regions, so 37 called the leaky-mode resonance (LMR) enhancement [9]. This 38 enhancement effect, however, is rather limited for photovoltaics 39 due to its restriction to resonance regions. On the other hand, the 40 off-resonance absorption enhancement has been known to be 41 more important than the resonance counterpart for photovoltaic 42 applications due to the large wavelength range of the solar 43 spectrum [1]. In this letter, we propose an effective light trap- 44 ping method in single semiconductor NWs by combining both 45 the LMR and the off-resonance absorption enhancements. We 46 have investigated this effective light trapping and correspond- 47 ing photocurrent enhancement in coaxial NWs that consist 48 of semiconductor NWs such as Si NWs (core) and coated 49 nonabsorbing dielectric materials such as SiO₂, Si₃N₄, and 50 Ta_2O_5 (shell). 51

II. THEORETICAL METHOD 52

As in the inset in Fig. 1(a), we have calculated the light 53 absorption of such coaxial NWs in the framework of the 54 Lorenz-Mie light scattering theory [10], where coaxial NWs 55 are treated as infinitely long cylinders, normally illuminated 56 by a plane wave with an incident propagation vector k_0 . The 57 cross sections of scattering (C_{sca}) and extinction (C_{ext}) for 58 transverse-electric (TE, electric field perpendicular to the axis 59 of the wire) and transverse-magnetic (TM, electric field parallel 60 to the axis) polarized lights are given by 61

$$C_{\text{sca}}^{\text{TE}} = \frac{4}{k_0} \left\{ \sum_{n=-\infty}^{\infty} |a_n|^2 \right\} \quad C_{\text{sca}}^{\text{TM}} = \frac{4}{k_0} \left\{ \sum_{n=-\infty}^{\infty} |b_n|^2 \right\} \quad (1a)$$
$$C_{\text{ext}}^{\text{TE}} = \frac{4}{k_0} \text{Re} \left\{ \sum_{n=-\infty}^{\infty} a_n \right\} \quad C_{\text{ext}}^{\text{TM}} = \frac{4}{k_0} \text{Re} \left\{ \sum_{n=-\infty}^{\infty} b_n \right\} \quad (1b)$$

where a_n and b_n are far-field scattering coefficients, and the 62 electric field amplitude E inside coaxial NWs can be readily 63 obtained by solving Maxwell's equations with the boundary 64 conditions at the core/shell and shell/air interfaces [10]. If the 65 incident light is unpolarized, like sunlight, we can express the 66 absorption cross section (C_{abs}) of coaxial NWs as [10] 67

$$C_{\rm abs} = \left(C_{\rm ext}^{\rm TE} + C_{\rm ext}^{\rm TM}\right)/2 - \left(C_{\rm sca}^{\rm TE} + C_{\rm sca}^{\rm TM}\right)/2.$$
(2)

Note that the volume absorption coefficient α_{abs} , a measure 68 of light absorption ability in coaxial NWs, can be given 69 as $C_{\rm abs}/V_{\rm core}$, where $V_{\rm core}$ is the core volume that is used 70 since the band gaps of the dielectric shells under investigation 71 (310-1100 nm) in this letter are over 4.0 eV (\sim 310 nm) so that 72 the shells have no contribution to the light absorption. It should 73 be noted that, although the light absorption is usually realized 74

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Fig. 1. (a) α_{abs} versus λ . (Inset) Schematic coaxial NW. Yellow (blue) stands for Si core (shell). m_0 (air), m_1 (shell), and m_2 (core) are the refractive indices. Thick arrows indicate incident light. See text for dashed boxes. (b)–(c) Cross-sectional $|E|^2/|E_0|^2$ distributions, shown only inside the core, at (b) off-resonance and (c) on-resonance: (Left) No shell and (right) SiO₂ shell of the same dimension as in (a).

75 along the axial direction in NW *array* solar cells [3], [4], we 76 consider only the normally incident light in this letter since 77 the light absorption practically occurs along the perpendicular 78 direction in *single* NW photovoltaic devices [5], [7].

III. RESULTS AND DISCUSSIONS

In Fig. 1(a), we show the wavelength (λ) dependence of 80 81 $\alpha_{\rm abs}$ in coaxial NWs with Si cores of radius r = 150 nm 82 and shells of thickness t = 30 nm for SiO₂ (refractive index 83 $m_1 = 1.5$), Si₃N₄ ($m_1 = 2.0$), and Ta₂O₅ ($m_1 = 2.3$) in air. 84 Without-shell α_{abs} is shown for comparison. The absorption 85 in the coaxial NWs is clearly enhanced at both resonance and 86 off-resonance regions for short $\lambda < \lambda_c \sim 480$ nm. However, 87 the latter enhancement is twice as much as the former: e.g., 88 24.6%, 51.7%, or 61.4% at off-resonance ($\lambda \sim 440$ nm) but 89 14.9%, 27.1%, or 29.5% at resonance ($\lambda \sim 450 \text{ nm}$) for SiO₂, 90 Si₃N₄, or Ta₂O₅, respectively. In contrast, the light absorption 91 appears to be comparable for long $\lambda > \lambda_c$, resulting in no 92 contribution to the photocurrent enhancement. Note that λ_c is 93 a characteristic wavelength, below which the light absorption 94 enhancement always occurs due to the shells, and can be 95 readily determined for given r, t, and m_1 , as found in the inset 96 in Fig. 2(e).

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97 The absorption behavior in Fig. 1(a) can be described in 98 terms of the field distribution inside coaxial NWs. Without 99 loss of generality, due to the similarity of TM and TE spectra 100 [5], we can consider the field (intensity) pattern only for TM. 101 The (normalized) field intensity can be defined as $|E|^2/|E_0|^2$, 102 where E_0 is the incident electric field amplitude. Note that the 103 light absorption in NWs α_{abs} is proportional to the product of 104 the field intensity and the imaginary part of the silicon refractive 105 index that roughly goes as $1/\lambda$ [10]. For $\lambda < \lambda_c$, the field 106 intensity at off-resonance is certainly increased in coaxial NWs 107 with SiO₂, as shown in Fig. 1(b), leading to the strong off-



Fig. 2. (a)–(d) α_{abs} versus λ and r. (e) I_V versus r, resulting from (a)–(d). (Inset) λ_c versus r. See text for circles and dashed lines.

resonance absorption enhancement previously discussed. Note 108 that, for this short-wavelength region, the resonance absorption 109 enhancement in with-shell NWs also results from the increase 110 AQ7 of the field intensity at resonance, but it is clearly smaller than 111 the off-resonance enhancement previously discussed. For $\lambda > 112$ AQ8 λ_c , although the field intensity of with-SiO₂ NWs is stronger 113 than that of without-shell NWs as shown in Fig. 1(c), the 114 resonance absorption in without-shell NWs is slightly bigger 115 than that in with-shell NWs, as can be seen at $\lambda \sim 710$ nm 116 in Fig. 1(a), since the resonance in without-shell NWs occurs 117 at the smaller wavelength than that in with-shell NWs. Note 118 AQ9 here that both off-resonance and resonance field intensities of 119 with- and without-shell NWs, as in Fig. 1(b) and (c), show the 120 same patterns that are typical modes inside Si NWs due to the 121 excitation of LMRs [5]. This can allow us to deduce that coaxial 122 NW (with shell) resonances also arise from the excitation of 123 LMRs. As a result, the light absorption in coaxial NWs (with 124 shell) can be enhanced at a desired wavelength by tuning the 125 core radius, likewise in Si NWs. 126

The absorption behavior in Fig. 1(a) can be understood by 127 means of the Fano effect [11] that is an interference effect 128 arising from the incident light and the localized reemitted LMR 129 light due to the core of subwavelength size in coaxial NWs. 130 The Fano interference effect is, however, of different origin 131 from that in conventional antireflection coating, where the in- 132 terference occurs due to the incident light and the phase-shifted 133 reflected light. Recently, the Fano effect has been observed 134 in spherical core/shell nanoparticles [12]. Note that, the lower 135 the field intensity, the weaker the LMRs. For strong LMRs 136 $(\lambda > \lambda_c)$, as in Fig. 1(c), such a thin shell (t = 30 nm) does not 137 play much role in the light absorption enhancement, since both 138 with- and without-shell absorptions are comparable. However, 139 for weak LMRs ($\lambda < \lambda_c$), as in Fig. 1(b), the incident light 140 $(\lambda \sim 430 \text{ nm})$ at given subwavelength scales (r = 150 nm and 141 t = 30 nm) appears to be localized via constructive interference 142 with the reemitted light, resulting in the strong enhancement of 143 the field intensity. 144

Furthermore, one can see the effects of the refractive index 145 m_1 on the absorption enhancement in coaxial NWs in Fig. 1(a). 146 The light absorption is significantly increased with increasing 147



Fig. 3. (a)–(c) I_V versus r and t. (d) Optimal t versus r, resulting from I_V in (a)–(c). (e) I_V versus r, corresponding to optimal t in (d). (f) I_V enhancement versus r, calculated from I_V in (e).

148 m_1 for $\lambda < \lambda_c$ due to the enhancement of the field intensity 149 inside the Si core with increasing m_1 . In addition, the size 150 of m_1 -driven red shifts of resonance peaks, as highlighted by 151 the dashed boxes, becomes large with increasing λ , resulting 152 from the increased phase shift of the interference between the 153 incident light and the reemitted light with increasing λ .

154 Now, we discuss the core size dependence of $\alpha_{\rm abs}$. In 155 Fig. 2(a)–(c), we present 2-D α_{abs} of coaxial NWs as a function 156 of λ and r for SiO₂ (t = 150 nm), Si₃N₄ (t = 110 nm), and 157 Ta₂O₅ (t = 100 nm), respectively. Note that we used different 158 shell thicknesses, selected from optimal regions that will be 159 clarified later. Fig. 2(d) also shows without-shell α_{abs} for 160 comparison. The absorption in coaxial NWs is clearly enhanced 161 at the off-resonance regions (see circles located at the same 162 spot). Note here that there are common characteristics between 163 with- and without-shell NWs: The number of resonant peaks is 164 augmented with increasing r [see the two dashed lines at 50 and 165 150 nm in Fig. 2(d)], and these resonant peaks clearly tend to 166 show substantial red shifts with increasing r (r-driven red shift). 167 These common characteristics further imply that the resonance 168 property of coaxial NWs is determined by the core radius.

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169 Under the standard solar spectrum AM1.5G, we can calculate 170 the photocurrent or short-circuit current per unit volume (I_V) as 171 $I_v(r, R) = q \int \Gamma(\lambda) \alpha_{abs}(\lambda, r, R) d\lambda$, where q is the elementary 172 charge and Γ is the photon flux density, and 100% collection 173 efficiency is assumed to evaluate the ultimate photocurrent. As 174 can be seen in Fig. 2(e), with-shell I_V is certainly increased for 175 the entire region, but $r \sim 10$ nm. The increased amount of I_V , 176 for example, at r = 24 nm is 25.7%, 48.4%, or 69.4% for SiO₂, 177 Si₃N₄, or Ta₂O₅, respectively. At $r \sim 10$ nm, the characteristic 178 wavelength λ_c (see inset) is so small ($\lambda_c \sim 370$ nm) that the 179 photocurrent can be hardly enhanced, since the light absorption 180 in the coaxial NWs is mainly enhanced for $\lambda < \lambda_c$, but the solar 181 irradiance is very weak for $\lambda < 370$ nm.

Finally, combining the effects of the thickness of shells (t)183 with those of the core size (r) and the refractive index of shells

 (m_1) on the light absorption, we present in Fig. 3(a)–(c) the 184 2-D I_V in coaxial NWs as a function of r and t for SiO₂, 185 Si_3N_4 , and Ta_2O_5 , respectively. All the contributions of r, t, 186 and m_1 to the photocurrent I_V fully emerge in these figures: 187 I_V periodically changes in the shell thickness, resulting from 188 the interference behavior of the Fano effect; I_V has maximal 189 values at the small core radius regions, consistent with [5]; 190 and I_V favors dielectric shells of a high refractive index, as 191 previously mentioned. Fig. 3(d) illustrates the optimal t as 192 a function of r, yielded from Fig. (a)–(c). In Fig. 3(e) and 193 (f), we present optimal I_V and its enhancement (the ratio of 194 with- to without-shell I_V) for different r's, corresponding to 195 the optimal t in Fig. 3(d). Except for the weak enhancement 196 of I_V around r = 10 nm, as previously discussed, a large 197 I_V enhancement is clearly observed in the coaxial NWs. The 198 maximum enhancement of I_V is 54.1%, 100.1%, or 102.2% 199 for the optimal core/shell NWs of Si $(r = 25 \text{ nm})/\text{SiO}_2$ $(t = 200 \text{ nm})/\text{SiO}_2$ 70 nm), Si $(r = 159 \text{ nm})/\text{Si}_3\text{N}_4$ (t = 180 nm), or Si (r = 201 nm)24 nm)/Ta₂O₅ (t = 124 nm), respectively. 202

We have demonstrated an effective light trapping method in 204 single coaxial NWs by coupling the off-resonance enhancement 205 with the LMR enhancement. From the light absorption calcula- 206 tions, we have found that the light absorption in coaxial NWs 207 can be significantly enhanced by tuning their core radius, shell 208 thickness, and refractive index. This strong enhancement read- 209 ily allowed us to obtain the photocurrent enhancement of up to 210 102%, implying that this effective light trapping technology can 211 be utilized for high-efficiency NW photovoltaic devices.

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